

Grain size reduction of SmFe₁₂-based powders; toward development of bulk high-performance permanent magnet

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Recent investigations on intrinsic magnetic properties of SmFe₁₂-based compounds with the ThMn₁₂ structure have shown that Sm(Fe_{0.8}Co_{0.2})₁₁Ti and Sm_{0.8}Zr_{0.2}(Fe_{0.8}Co_{0.2})_{11.5}Ti_{0.5} alloy compounds have comparable intrinsic hard magnetic properties and better temperature dependence with those of Nd₂Fe₁₄B [1]. The remaining challenge is to further develop those intrinsic properties into practically useful extrinsic ones, particularly coercivity. In this work, we focused on reduction of the grain size of SmFe₁₂-based compounds by jet-milling and investigated the phase stability of the powders with different particle size [2,3]. We also introduce a few grain boundary phases that can exchange decouple SmFe₁₂-based grains in order to obtain high coercivity.

Starting alloys with compositions of Sm(Fe_{0.8}Co_{0.2})₁₁Ti, Sm(Fe_{0.8}Co_{0.2})_{10.5}Ga_{0.5}Ti, and Sm(Fe_{0.8}Co_{0.2})_{10.5}Cu_{0.5}Ti were prepared by induction melting. The ingots were crushed into coarse powders and hydrogen decrepitated to reduce the particle size below 100 μm. The powders were jet-milled for the particle refinement study. The magnetic properties were measured using a SQUID-VSM. Microstructure of the samples were analyzed using SEM/FIB (Carl Zeiss 1540EsB) and TEM (Titan G2 80-200).

We used three different compositions of SmFe₁₂-based alloys and demonstrated fine, anisotropic, single-crystalline SmFe₁₁Ti-based micro-particles with high roundness by jet-milling using N₂ gas. The smallest particle size of 2.7±0.6 μm and roundness of 0.73±0.13 was achieved in the case of Sm(Fe_{0.8}Co_{0.2})_{10.5}Cu_{0.5}Ti, as shown in Fig. 1 (a). Although the ThMn₁₂-type structure is preserved even at high milling gas pressures, the coercivity of the powders remained below 0.2 T. The intergranular phase that originally exist in the as-cast alloy is removed or changed to an amorphous surface phase as shown in Fig. 1 (b). Detailed characterization of jet-milled powders using BF and HAADF-STEM showed that the SmFe₁₂-based jet-milled powders experience plastic deformation through crystallographic slip and the formation of slip bands during jet-milling process (Fig. 1(c)). Observed projection of the atoms in HAADF-STEM image (Fig. 1(c)) show that Sm atoms are missing in the slip bands

indicating the crystal structure is damaged at the slip bands. These defects decrease local magnetocrystalline anisotropy compared to the matrix region that can be nucleation centers for the formation of reverse magnetic domains and thus limiting the coercivity. We will show that possible solutions for minimizing defect formation are, using lower milling gas pressure, performing a post-annealing step or using Ga doped starting alloy.

Temperature-dependent XRD results showed that unlike large sized powders (>100μm), jet-milled powders with a size of smaller than 5μm decompose from the surface at temperatures above 600°C resulting in formation of α-Fe. We will also show low-melting alloys with good wettability with SmFe₁₂-powders that can be used for metal bonded magnets.

Reference

[1] P. Tozman, H. Sepehri-Amin *et al.* Acta Mater. 153 (2018) 534.

[2] I. Dirba, H. Sepehri-Amin *et al.* Acta Mater. 165 (2019) 9. [3] I. Dirba, H. Sepehri-Amin *et al.* Submitted.

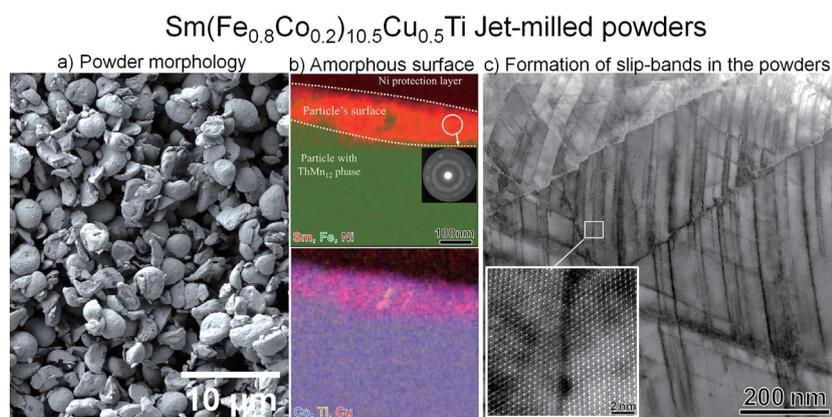


Figure 1: (a) Secondary electron (SE) SEM image showing morphology of jet-milled powders, (b) STEM-EDS maps and micro-diffraction pattern showing amorphous Sm-Cu rich phase at the surface of jet-milled powders, and (c) bright field (BF)-STEM and high resolution HAADF-STEM image (inset) obtained from inside of the jet-milled powders with composition of Sm(Fe_{0.8}Co_{0.2})_{10.5}Cu_{0.5}Ti.