

## Large perpendicular magnetic anisotropy in Fe/MgAl<sub>2</sub>O<sub>4</sub> heterostructures

Qingyi Xiang<sup>1,2</sup>, Ruma Mandal<sup>2</sup>, Hiroaki Sukegawa<sup>2</sup>, Yukiko K. Takahashi<sup>2</sup> and Seiji Mitani<sup>1,2</sup>

<sup>1</sup>Graduate School of Pure and Applied Sciences, University of Tsukuba, Tsukuba 305-8577, Japan

<sup>2</sup>National Institute for Materials Science, Tsukuba 305-0047, Japan

MgAl<sub>2</sub>O<sub>4</sub> is considered a promising alternative barrier material to MgO for magnetic tunnel junctions (MTJs) due to its tunable lattice constant<sup>1</sup>. The interface-induced perpendicular magnetic anisotropy (PMA) at an MgAl<sub>2</sub>O<sub>4</sub> interface is a crucial property for applications of perpendicularly magnetized MTJs (p-MTJs). Based on a recent theoretical calculation,<sup>2</sup> the areal PMA energy density of  $\sim 1.3$  mJ/m<sup>2</sup> was predicted at an Fe/MgAl<sub>2</sub>O<sub>4</sub>(001) interface, which is nearly comparable to that at an Fe/MgO(001) interface ( $\sim 1.5$ – $1.7$  mJ/m<sup>2</sup>). However, a much smaller PMA energy density  $\sim 0.4$  MJ/m<sup>3</sup>,<sup>3</sup> comparing with  $\sim 1.4$  MJ/m<sup>3</sup> in Fe/MgO,<sup>4</sup> has been experimentally reported in Fe/MgAl<sub>2</sub>O<sub>4</sub>(001) where the MgAl<sub>2</sub>O<sub>4</sub> layers were prepared by post-oxidization of an Mg-Al metallic layer. Therefore, further improvement in the PMA energy of ultrathin-Fe/MgAl<sub>2</sub>O<sub>4</sub>(001) interfaces is expected if a sharp interface is obtained by suppressing atomic intermixing and over-oxidation through process optimization. In this study, we report achievement of large PMA at an Fe/MgAl<sub>2</sub>O<sub>4</sub> by introducing electron-beam deposition of MgAl<sub>2</sub>O<sub>4</sub>.<sup>5</sup>

Stacks of Cr buffer(30)/Fe(0.7)/MgAl<sub>2</sub>O<sub>4</sub>( $t_{\text{MAO}} = 2$  or 3 nm) (unit in nm) multilayers were epitaxially grown on an monocrystalline MgO(001) substrate by electron-beam evaporation. The Cr and Fe were post-annealed at 800°C and 250°C, respectively. The MgAl<sub>2</sub>O<sub>4</sub> were post-annealed at various temperatures between 350°C and 500°C to modify the interface conditions. Magnetic properties were investigated using a vibrating sample magnetometer(VSM) and VSM incorporated with superconducting quantum interference device (SQUID). The ultrafast magnetization dynamics property was measured by the time-resolved magneto-optical Kerr effect (TR-MOKE) method.

Figure 1 shows the  $M$ - $H$  curve of an optimized Fe (0.7 nm)/MgAl<sub>2</sub>O<sub>4</sub>( $t_{\text{MAO}} = 2$  or 3 nm) interface with a large PMA energy up to  $\sim 1.0$  MJ/m<sup>3</sup>, comparable to the reported value for an Fe (0.7 nm)/MgO<sup>4</sup> ( $\sim 1.4$  MJ/m<sup>3</sup>). We also found that the PMA energy and saturation magnetization ( $M_s$ ) were not very sensitive to measurement temperature, where from 100K to 300K,  $K_i$  drops from  $\sim 2.0$  mJ/m<sup>2</sup> to  $\sim 1.7$  mJ/m<sup>2</sup>. The effective damping constant was also evaluated to be  $\sim 0.02$  by TR-MOKE under high magnetic fields. This study demonstrated robust interface PMA in ultrathin-Fe/MgAl<sub>2</sub>O<sub>4</sub>, which is useful for p-MTJ applications.

This study was partly supported by the ImPACT program of the Council for Science, Technology and Innovation (Cabinet Office, Government of Japan) and JSPS KAKENHI Grant Number 16H06332.

### Reference

- 1) H.Sukegawa, H. Xiu, T. Ohkubo, T. Furubayashi, T. Niizeki, W. Wang, S. Kasai, S. Mitani, K. Inomata, and K. Hono: *Appl. Phys. Lett.* **96**, 212505(2010)
- 2) K. Masuda and Y. Miura, *ArXiv*:1803.10428 (2018).
- 3) J. Koo, H. Sukegawa, and S. Mitani, *Phys. Status Solidi RRL* **8**, 841 (2014).
- 4) J.W. Koo, S. Mitani, T.T. Sasaki, H. Sukegawa, Z.C. Wen, T. Ohkubo, T. Niizeki, K. Inomata, and K. Hono, *Appl. Phys. Lett.* **103**, 192401 (2013).
- 5) Q.Xiang, R. Mandal, H. Sukegawa, Y.K. Takahashi and S.Mitani, *Appl. Phys. Express.* **11**, 063008. (2018)

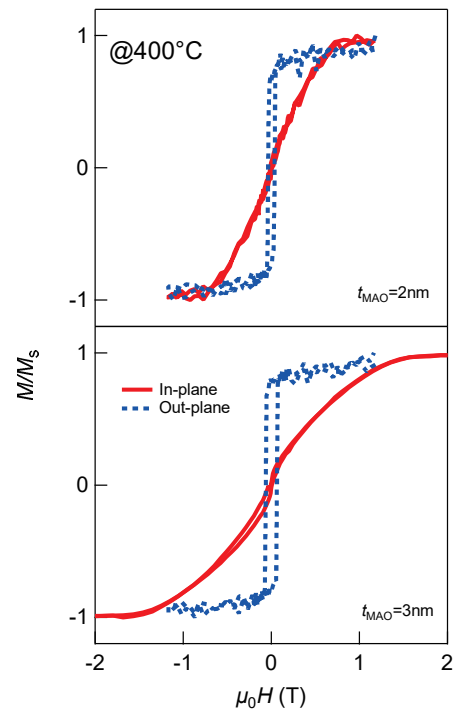


Figure 1 M-H curves for samples annealed at 400°C with  $t_{\text{MAO}} = 2$  and 3 nm.