

## Improvement of MR output in $\text{Co}_2\text{MnGa}_{0.25}\text{Ge}_{0.75}$ Heusler alloy based CPP-GMR devices through composition tuning

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Many recent investigations have shown that the usage of Heusler alloy is an effective way to enhance the MR output in current-perpendicular-to-plane giant magnetoresistance (CPP-GMR) devices due to its half-metallic nature as predicted by band structure calculations. We have developed highly spin polarized  $\text{Co}_2\text{MnGa}_{0.25}\text{Ge}_{0.75}$  full Heusler alloy and achieved large MR output by using it as ferromagnetic electrode in CPP-GMR devices [1]. In our previous study,  $\text{DO}_3$ -disorder was found with annealing temperature ( $T_a$ ) higher than 450 °C in the  $\text{CoMnGaGe}$  film with the composition of  $\text{Co}_{52.7}\text{Mn}_{25.6}\text{Ga}_{6.1}\text{Ge}_{16.5}$ . The  $\text{DO}_3$ -disorder (Co antics) is thought to decrease the spin polarization of Co-based Heusler alloy based on the band structure calculations [2]. MR output was thus degraded due to the appearance of Co antics in our previously study using  $\text{CoMnGaGe}$  film. In this work, we report an improvement of MR output in CPP-GMR device by suppressing the Co antics through the composition tuning in  $\text{CoMnGaGe}$  Heusler alloy.

Epitaxial  $\text{CoMnGaGe}$  thin films with the composition of  $\text{Co}_{46.4}\text{Mn}_{28.8}\text{Ga}_{7.1}\text{Ge}_{17.7}$  (refer to Mn-rich CMGG) or  $\text{Co}_{52.7}\text{Mn}_{25.6}\text{Ga}_{6.1}\text{Ge}_{16.5}$  (refer to Co-rich CMGG) were prepared by using  $\text{MgO}(100)$  single crystal substrate in ultra-high vacuum magneto-sputtering system. CPP-GMR devices were fabricate through conventional EB lithography and Ar ion milling process. Anisotropy magneto resistance (AMR) measurement was performed in a physics property measurement system (PPMS) and GMR measurement was performed in the CPP-geometry.

Fig. 1 shows AMR as function of  $T_a$  in Mn-rich CMGG film and Co-rich CMGG films with thickness of 50 nm, deposited on  $\text{MgO}$  substrate. The amplitude of negative AMR reaches maximum at  $T_a=500$  °C, and then decreases with further increasing  $T_a$ . This can be interpreted by the reduction of spin polarization caused by Co antics at high  $T_a$  in Co-rich CMGG film, which has been discussed in our previous work [3]. On the other hand, AMR in Mn-rich CMGG film shows monotonic relation with  $T_a$  up to 650 °C, which might indicate suppressed Co antics in Mn-rich CMGG film by increasing the concentration of Mn. Fig. 2 shows CPP-GMR output as function of  $T_a$  in Mn-rich CMGG film and Co-rich CMGG films based CPP-GMR devices. The Mn-rich CMGG film based CPP-GMR devices show more robust dependence of  $T_a$ . It is gives strong evidence of suppressed Co antics in CMGG film thus leading to lager spin polarization. The  $\Delta RA$  drops at high  $T_a$  might be due to the chemical diffusion between Mn and Ag in case of Mn-rich CMGG film based CPP-GMR devices.

[1]Y. K. Takahashi et al, J. Appl. Phys. **113**, 1223901 (2013) [2]S. Picozzi et al, J. Phys.: Condens. Matter, **19**, 315215 (2007) [3]Y. Sakuraba et al, Appl. Phys. Lett. **104**, 172407 (2011)

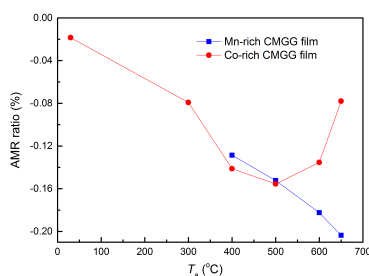


Fig. 1 AMR as function of  $T_a$  in Mn-rich (blue square) and Co-rich CMGG (red circle) films.

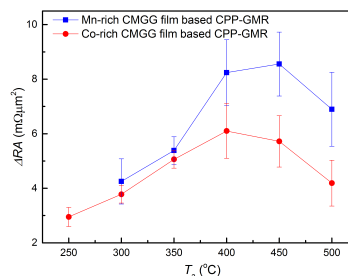


Fig. 2  $\Delta RA$  as function of  $T_a$  in Mn-rich (blue square) and Co-rich CMGG (red circle) films based CPP-GMR devices.